# $C_2H_5OH$ and LPG sensing properties of $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> microflowers prepared by hydrothermal route

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## ABSTRACT

The flower-like micron-structure of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> was synthesized via hydrothermal treatment at 140 °C for 24 h using  $Fe(NO_3)_3.9H_2O$  and Na<sub>2</sub>SO<sub>4</sub> as the precursors. A thin film constructed by the as-prepared material was created by spin coating technique. The structure, morphology, and composition of the samples were characterized by X-ray diffraction (XRD), field emission scanning electron microscopy (FESEM). The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> microflowers (MFs) with average diameter of several micrometers are assembled of nanorods which possess average diameter and length of 40 nm and hundred nm,

respectively. The gas sensing properties of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> film were tested with ethanol (C<sub>2</sub>H<sub>5</sub>OH) and liquefied petroleum gas (LPG) at the operating temperatures of 225–400 °C. The sensor response of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> film reached highest sensitivity to C<sub>2</sub>H<sub>5</sub>OH and LPG at 275 °C and 350 °C, respectively. The thin film exhibited higher sensitivity and lower working temperature to C<sub>2</sub>H<sub>5</sub>OH than those to LPG. The film can detect minimum concentration of 250 ppm C<sub>2</sub>H<sub>5</sub>OH. The response time of the film to C<sub>2</sub>H<sub>5</sub>OH is approximately 30 s.

Keywords:  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, gas sensor, microflower, nanorod, hydrothermal.

### 1. INTRODUCTION

Hematite ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>) is the most stable iron oxide under ambient conditions which behaves as an n-type semiconducting material with band gap of 2.2 eV [1]. It is frequently applied as semiconducting material, dielectric material, magnetic material, sensitive material and catalyst, etc. [2-6]. In recent years, much effort has been focused on the fabrication of nanostructure materials with a desired size, morphology and porosity, owing to their special electrical, optical, magnetic, and physical/chemical properties that are superior to those bulk materials [7-10]. Stimulated by both the promising applications of iron oxide and the novel properties of nanoscale materials, many synthesized scientists have a-Fe<sub>2</sub>O<sub>3</sub> nanostructured materials in various geometrical morphologies such nanotubes as [8], nanoparticles [11], nanowires and nanobelts [12], nanorods [1, 10, 13], nanocubes, sea urchin-like [14], nanoplates [15], etc. Ferric oxide has been prepared by liquid-phase deposition method (LPD) [16], plasma enhanced chemical vapor deposition (PECVD) [17], ion-sputtering [18], ultrasonic spray pyrolysis [19], sol-gel route [2], hydrothermal method [1, 11, 13, 14], etc. Among those methods, hydrothermal treatment is a simple and reliable method for synthesizing nanostructures with designed chemical components and controlled morphologies. Recently, three-dimensional (3D) superstructures assembled with one-dimensional nanorods have attracted much attention due to their higher specific surface area [20].

In this study, we report a facile route to synthesize  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> MFs without any surfactant and template via a low temperature (140 °C) hydrothermal approach. The material can be fabricated with large scale and good reproducibility. Besides, the ethanol and LPG sensing characteristics of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> film are also investigated. The results indicated that the sensor response of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> film reached highest sensitivity to ethanol vapor and LPG at operating temperature of 275 °C and 350 °C, respectively.

### 2. EXPERIMENTAL

The preparation process of flower-like  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanostructures is introduced in Fig.1. In a typical synthesis, 100 ml 0.075 M sodium sulfate (Na<sub>2</sub>SO<sub>4</sub>) solution was added to 100 ml 0.075 M iron (III) nitrate (Fe(NO<sub>3</sub>)<sub>3</sub>) solution. After stirring for 30 min, 10 ml deionized water was

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added to form a homogeneous solution. The mixed solution was sealed into a Teflon-lined stainless steel autoclave of 50 ml capacity and heated at 140 °C for 24 h. After treatment, the autoclave was cooled to room temperature naturally. The red-brown powder was isolated by centrifugation, washed by deionized water and absolute ethanol several times, and finally dried at 80 °C for 24 h in air. The obtained powder was then characterized by XRD (Bruker D8 Advance X-ray diffractometer, Germany) and scanning electron microscopy (Hitachi S4800, Japan). The a-Fe<sub>2</sub>O<sub>3</sub> powder was mixed and grinded with water and PEG to form a gas-sensing paste. The α-Fe<sub>2</sub>O<sub>3</sub> material was coated on silicon substrate deposited interdigitated platinum electrodes. In order to improve their stability and repeatability, the gas sensor was annealed at 600 °C for 2 h in air. The gas sensing properties of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> film were tested to C<sub>2</sub>H<sub>5</sub>OH (250-2000 ppm) and LPG (2500-10000 ppm) at operating temperatures of 225-400 °C.

### 3. RESULTS AND DISCUSSION

In order to identify whether there is any influence of thermal treatment at 600 °C for 2 h on morphologies and crystal structures of a-Fe<sub>2</sub>O<sub>3</sub> mircroflowers (MFs), we measured XRD spectra and took the SEM images of the α-Fe<sub>2</sub>O<sub>3</sub> MRs before and after annealing, and these results are demonstrated in Figs. 2 and 3. The main diffraction peaks of the freshly-obtained Fe<sub>2</sub>O<sub>3</sub> MRs can be well indexed to a rhombohedral  $Fe_2O_3$  with lattice parameters of a=b=5.0016 Å, c=13.6202 Å,  $\alpha = \beta = 90^{\circ}$ ,  $\gamma = 120^{\circ}$ , the space group is R-3c (Fig. 2a). Fig. 2b shows the X-ray diffraction (XRD) pattern of the obtained a-Fe<sub>2</sub>O<sub>3</sub> after annealing at 600 °C for 2 h. All the reflection peaks in the XRD pattern are indexed to the single crystal of hexagonal structure of Fe<sub>2</sub>O<sub>3</sub> with lattice constants of a=b=5.038 Å, c=13.772 Å and  $\alpha$ = $\beta$ =90°,  $\gamma$ =120°, the space group is R32/c (JCPDS Card No. 33-0664). Thus, the crystal structure of Fe<sub>2</sub>O<sub>3</sub> nanorods transfers from rhombohedral before annealing into hexagonal phase after annealing. No further peaks of another phases was observed, suggesting that the product was high purity. The strong and narrow diffraction peaks observed in the pattern indicate that the material possesses a good crystallinity. In the XRD pattern, the (104) diffraction peak has the strongest reflection, indicating that the (104) is the preferential growth plane of the nanorods. The crystallite size of  $Fe_2O_3$  nanorods are estimated using the Scherrer equation based on the (104) peak, and it is found to be around 30 nm before and after annealing.



Figure 1. Scheme of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> synthesis.



Figure 2. XRD pattern of as-prepared (a)  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> microflowers and after annealing at 600 °C for 2 h (b).



Figure 3. SEM images of as-prepared  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> MFs with magnification of 10k (a), 100k (b) and 600 °C-annealed  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> MFs with magnification of 100k (c).

The SEM image of the as-prepared sample at magnification of 10k (or ×10000 times) in Fig. 3a shows the microflowers with diameters of 2-3  $\mu$ m. The microflowers are constructed by wrapped layer of oriented nanorods with diameters of 30–50 nm and lengths of 200–300 nm (Fig. 3b). It can be seen that the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanorods are arranged in an orderly fashion. Fig. 3c shows the SEM image of the sample after heat treatment at 600 °C for 2 h. It was observed that the rod-like morphology was maintained, both diameter and length of the nanorods were similar to those of as-prepared product but the rod surface seems smoother.

Figure 4 shows the gas-sensing characteristics of the α-Fe<sub>2</sub>O<sub>3</sub> MFs in response to  $C_2H_5OH$ . It is known that the sensing characteristic of α-Fe<sub>2</sub>O<sub>3</sub> for a special gas is usually dependent on the temperature, so parallel experiments were carried out in the range of 225-325 °C to optimize the proper working temperature of the sensor. As is shown in Fig. 4a, the results indicated that sensor showed the highest response to C<sub>2</sub>H<sub>5</sub>OH at 275 °C. Regarding to the complex morphology, the thin film exhibits surface roughness which may provide many sites to adsorb the gas molecules, therefore enhances the sensitive properties.

desorption of gas molecules on the surface of the sensing structure [2]. It is possibly related to the chemical reaction kinetics between gas molecules and oxygen ions adsorbed on the surface of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> superstructures. The relatively looser bundle aggregates can act as gas diffusion channels making the diffusion much easier. The surface-to-volume ratio is relatively high as a result of small diameter of nanorods which enables the gases to access all surfaces of the nanorods contained in the sensing unit. Thus, it is reasonable to believe that sensor made with aligned  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanorods should have enhanced sensitivity. The response of the sensor based on the porous flower-like α-Fe<sub>2</sub>O<sub>3</sub> nanostructures is much higher than that of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles under the same condition [7] because the porous flower-like nanostructure may possess high surface area which can provide more adsorption-desorption sites for gas molecules compared to that of the nanoparticles. The pseudo-cubic shaped  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> particles with the mean size of about 58 nm showed high sensitivity toward C<sub>2</sub>H<sub>5</sub>OH. Its response to 50 ppm C<sub>2</sub>H<sub>5</sub>OH at room temperature was 19 [2]. The response of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanotubes to 50 ppm

It is generally accepted that the change in resistance is mainly caused by the adsorption and

C<sub>2</sub>H<sub>5</sub>OH at room temperature was 26, about five times greater than that of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles [8]. The response of the hollow sea urchin-like  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanostructures to 42 ppm ethanol at 350 °C was 7.2, which was excess twice that of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanocubes [14]. The response of porous  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanorods to 1000 ppm C<sub>2</sub>H<sub>5</sub>OH at 250 °C was 175, which was almost several decade times greater than that of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles under the same ethanol concentration [10].





The chemical reaction rate is slow at lower temperature, leading to a lower response of the sensor. When the operating temperature is high (above 275 °C), desorption process becomes dominant, higher temperature hampers the diffusion of tested gases towards the sensing surface resulting in lowering the diffusion length, leading to reducing of response to ethanol. We thus select 275 °C as the proper working temperature to proceed with the subsequent detections. Fig. 4b illustrates a typical responserecovery characteristic of the sensor based on the porous flower-like α-Fe<sub>2</sub>O<sub>3</sub> microstructure to C<sub>2</sub>H<sub>5</sub>OH with concentrations of 250, 500, 1000, 1500 and 2000 ppm at 275 °C. It can be seen that the response of the sensor increases dramatically with the increase in the ethanol concentration and the highest response is 18 to 2000 ppm ethanol. After several cycles, the resistance of the sensor can recover its initial states, which indicates that the sensor has good reversibility.

Fig. 4c is the plot of sensitivity versus the concentration of  $C_2H_5OH$ . The sensitivity increases linearly to the ethanol concentration from 250 to 2000 ppm. The linear relationship between the sensitivity and the ethanol concentration was also observed in the previous reports [2]. The response and recovery times towards 2000 ppm  $C_2H_5OH$  are 30 s and 460 s, respectively. Such behavior can be understood by considering the dependence of oxygen adsorption on the operating temperature of the sensor. In the ambience air, the state of oxygen adsorbed on the surface of the material undergoes the following reactions. Oxygen in air is adsorbed onto material surface:

$$O_2(gas) \leftrightarrow O_2(ads) \tag{1}$$

Then, the adsorbed oxygen changes to ion  $O_{2}^{-}$  following the reaction:

$$O_2(ads) + e^- \leftrightarrow O_2^-(ads)$$
 (2)

At high temperature, the ions  $O_2^-$  change to ions  $O^-$ :

$$O_2^-(ads) + e^- \leftrightarrow 2O^-(ads)$$
 (3)

where (gas) and (ads) denote gas phase and adsorbed species. The oxygen species capture electrons from the material, leading a decrease in electron concentration. When the target gas was injected in the test chamber and reacted with the adsorbed oxygen, electrons traped by the adsorptive states can be released into the conduction band, which resulted in a decrease in sensor resistance. Ethanol reacts with the adsorbed oxygen according to following reaction:

$$C_2H_5OH + 6O^- \rightarrow 2CO_2 + 3H_2O + 6e^- \quad (4)$$

In Fig. 4d, we examine the sensitivity to LPG and the results show that the optimal operating temperature obtained is different from the result shown in Figure 4a. In this experiment, the sensitivity of the sensor increases with increasing operating temperature and reaches its maximum at 350 °C. The maximum response to 10000 ppm LPG at 350 °C is only 3.8, which is about five times lower than that to 2000 ppm ethanol. This behavior may relate to the differences of electron donating ability between C<sub>2</sub>H<sub>5</sub>OH and LPG, in which C<sub>2</sub>H<sub>5</sub>OH possesses a higher value due to the high electronegativity of carbon comparing with lower atom electronegativity of oxygen atom. The overall

$$C_n H_{2n+2} + 2O^- \rightarrow H_2 O + C_n H_{2n} O^- + e^-$$
 (5)

Fig. 4f shows the linear relationship between the sensitivity and the LPG concentration of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> MFs. The response and recovery times of the thin film to 10000 ppm LPG are 30 s and 335 s, respectively. It means that this film performs quick response and long recovery duration toward both ethanol and LPG. Furthermore, it is found in the sensing output that the measurement circle is well repeatable without major change in the baseline resistance.

### 4. CONCLUSION

In summary, flower-like  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> microstructures were synthesized by a simple hydrothermal treatment at 140 °C for 24 h. The

 $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> MFs were composed of regular nanorods with average diameter of 40 nm and average length of hundreds nm. Furthermore, the gas-sensing measurements demonstrated that the sensors based on the porous flower-like  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> exhibited good sensitivity to C<sub>2</sub>H<sub>5</sub>OH. The sensor response was 18 towards 2000 ppm of C<sub>2</sub>H<sub>5</sub>OH at 275 °C. The sensor response to 10000 ppm LPG was 3.8 at 350 °C. This sensor showed a linear, stable and reproducible response to C<sub>2</sub>H<sub>5</sub>OH in the range of 250–2000 ppm, without significant baseline resistance shift during the test. Furthermore, it exhibits quick response to the C<sub>2</sub>H<sub>5</sub>OH (30 s).

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# Tính chất nhạy khí C<sub>2</sub>H<sub>5</sub>OH và LPG của hoa micro α-Fe<sub>2</sub>O<sub>3</sub> chế tạo bằng phương pháp thủy nhiệt

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# TÓM TẮT

Cấu trúc micro hình hoa  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> được tổng hợp bằng xử lí thủy nhiệt ở 140 °C trong 24 h sử dụng tiền chất Fe(NO<sub>3</sub>)<sub>3</sub>.9H<sub>2</sub>O và Na<sub>2</sub>SO<sub>4</sub>. Màng mỏng tạo bởi vật liệu này được tạo bằng kĩ thuật quay phủ. Cấu trúc, hình thái và thành phần của mẫu được xác định bởi giản đồ nhiễu xạ tia X (XRD), hiển vi điện tử quét phát xạ trường (FESEM). Hoa micro  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> có đường kính trung bình khoảng vài µm được sắp xếp bởi các thanh nano có đường kính trung bình khoảng 40 nm và chiều dài hàng trăm nm. Tính chất nhạy khí của màng  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> được kiểm tra với hơi ethanol (C<sub>2</sub>H<sub>5</sub>OH) và khí ga hóa lỏng (LPG) ở nhiệt độ làm việc trong khoảng 225 đến 400 °C. Độ đáp ứng của màng  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> đạt cực đại với C<sub>2</sub>H<sub>5</sub>OH và LPG tương ứng ở 275 °C và 350 °C. Mẫu cho thấy độ nhạy cao hơn và nhiệt độ làm việc thấp hơn với C<sub>2</sub>H<sub>5</sub>OH so với LPG. Màng có thể phát hiện nồng độ nhỏ nhất của C<sub>2</sub>H<sub>5</sub>OH là 250 ppm. Thời gian đáp ứng của màng với C<sub>2</sub>H<sub>5</sub>OH xấp xỉ 30 s.

Từ khóa:  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, cảm biến khí, hoa micro, thanh nano, thủy nhiệt.

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